

## Effect of hydroxyurea (HU) on gelatinization mechanism of type I collagen suspensions

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### ABSTRACT

Mutant hemoglobins thermodynamic phase transitions (HbS, HbC, etc.) have been the causes of serious hemoglobinopathies that have been affecting mankind since the dawn of human race. HbS polymerization is the best known of these transitions and the original cause of sickle cell anemia. Hydroxyurea (HU,  $\text{NH}_2\text{CONHOH}$ ) is widely used as the main drug administered for the treatment of Sickle Cell Anemia (SCA). Several hypotheses have been raised on the effects of HU on SCA. The most widespread and accepted one points to the production of fetal hemoglobin by the bone marrow, hindering the polymerization of deoxygenated HbS. Moreover, recent studies have reported that the short term in vivo effects of the HU have not been elucidated. By ordinary techniques of Materials Science (scanning electron microscopy, viscoelasticity, viscosity measurements, DSC and weight loss) we assessed the anti-nucleating, depolymerizing, and anti-gelling direct effects of HU on suspensions and type I bovine collagen gels. Scanning electron microscopy (SEM) revealed that the addition of HU results in the disappearance of the percolating three-dimensional network typical of collagen gels. Vane Rheometry tests pointed to a reduction in the yield stress ( $\sigma_0$ ) by adding HU into collagen gels with a concentration of 2.2 % w/v. In the liquid phase (sol), suspensions increased viscosity over time but decreased with the addition of HU, highlighting the depolymerizing effect of this substance. The weight loss of the sols is also affected by the addition of HU probably related to the lowering of the interactions between collagen/water as HU is added. The DSC essays indicated changes in the Cp curves as HU is added to the gel, decreasing the structural relaxations commonly observed in those gels. As a result, HU showed an effect on the phase transition of the studied gel, strengthening the hypothesis of anti-nucleating and anti-gelling action thereof on physical, reversible gels, and possibly on physical, reversible gels of mutant hemoglobin (HbS).

**Keywords:** hydroxyurea, gelation, collagen, polymerization, sickle cell anemia.

### 1. INTRODUCTION

First synthesized in the XIX century [1] and still widely used hydroxyurea (HU,  $\text{NH}_2\text{CONHOH}$ ) is a substance utilized in such diverse human activities ranging from the nuclear industry [2] to Medicine [3]. This remarkable compound [4] is the main drug administered for the treatment of sickle cell anemia since the beginning of the 21<sup>st</sup> century. Recently it has been reported that HU occurs naturally in invertebrates (mollusks and crustaceans), fishes, amphibians and mammals [5]. Various studies show it has antiviral, antibacterial, and antineoplastic properties [6].

Sickle cell anemia is a hemoglobinopathy in which erythrocytes assume a "sickle" shape [7, 8] upon mutant hemoglobin S (HbS) suspension solidification (gelling). Human red blood cells show hemoglobin concentrations of nearly 150 g/l and the World Health Organization (WHO) recommends standard cut-off points at 130 g/L for men and 120g/L for women [9]. Normal hemoglobin molecules do not polymerize [10], so the transition between concentrated suspensions to gel is absent; hence, the impressive mechanical deformability and viscoelasticity of erythrocytes [11] are maintained. In mutant HbS, the substitution of valine for glutamic acid in the hemoglobin [12,13] molecule causes HbS intracellular polymerization when at deoxygenated state and, consequently, the sickling of erythrocytes occurs. The loss of suppleness and deformability, typical of a normal Red Blood Cell (RBC) filled with hemoglobin HbA suspension, is caused by sickling, altering blood stream (collisions of hardened red blood cells in the endothelium) [14] and oxygen supply,

thus causing serious damage to vital organs with long and painful episodes, which are among of the most distressing aspects of the disease. According to data from "Neonatal heel prick" tests applied during the National Neonatal Screening Program - PNTN (Ministry of Health of Brazil, 2012), nearly 3,500 Brazilian children were born with sickle cell anemia per year, and about 200,000 infants had the "trait", i.e. carrying one mutated gene from one of the parents but without developing the symptoms [15].

Although it was the first genetic disorder described worldwide by Linus Pauling in 1947 [8], the major medical advance was the use of hydroxyurea [16,17, 18], which is today strongly recommended by the international medical community. Previous studies reported the effect of HU as an increase in fetal hemoglobin (HbF) levels and, in turn, amelioration in symptoms [19,20,21]. HU may have a direct effect on the HbS polymer [22,23,24]. The hypothesis of HU anti-gelling effect [25,26] advocated in this paper was evaluated in type-I collagen gels. Collagen [27] is the most abundant protein in mammals, a fibrous protein found in skin, bones, cartilage, and connective tissues, and forms gels of the same nature as deoxygenated HbS gels, that is, physical reversible gels.

The long-term goal pursued in this study, and in ongoing studies, is to generate methods for the testing of HbS gels, aiming at new substances for the treatment of the disease.

## 2. MATERIALS AND METHODS

Collagen was solubilized in 0.02M acetic acid solution (at pH 2.0) to a final concentration of  $6.7 \pm 0.5\%$ . After, the solution was heated in a thermostatic bath (TC-500, Brookfield) at  $60^\circ\text{C}$  for 60 minutes; then, a sample (C1: 660g) was centrifuged at 2500-rpm rotation for 15 minutes (Mikro 220R, Hettich Zentrifugen) at  $40^\circ\text{C}$ . The pH was adjusted to 7.4 with a 0.05M NaOH solution. The hydroxyurea (HU 98%, M/S Sigma-Aldrich) doses utilized in the samples were named according to Table 1. For the Vane Rheometry (see below item 2.2) the dose was based on the recommendation established by the Brazilian Ministry of Health, i.e., 35 mg per kg considering an adult weighing 70 kg with a blood volume of 5 liters [28]. Two gel types were prepared: one heterogeneous ( $6.7 \pm 0.5\%$  w/v) and another translucent and homogeneous ( $2.2\% \pm 0.5\%$ ).

**Table 1:** Identification of collagen gels

Sample code	%(Collagen Type I /Water)	HU concentration on water (mg/ml)	HU content compared to the maximum human dose per day [28]	Gel characteristics (Heterogeneous/opaque or homogeneous/ translucent)
C1	6.7	0	0	opaque
C2		20	40x	opaque
C3		40	80x	opaque
Cs1	2.2	0	0	translucent
Cs2		0.49	1x	translucent
Cs3		0.98	2x	translucent
Cs4		20	40x	translucent
Cs5		40	80x	translucent

### 2.1 Scanning electron microscopy (SEM)

Samples were prepared following the protocol described by Souza [29]. First, the gels were placed on circular glass slides, previously immersed in a polylysine solution to enhance adhesion. Second, these materials were fixed with 2.5% glutaraldehyde solution, in 0.1M cacodylate buffer (primary fixative), for 1 hour, and then washed 3 times for 10 min in 0.1 M cacodylate buffer. Third, they were immersed in secondary fixative (1.0% osmium tetroxide) for 1 hour, being then washed 3 times for 10 min in 0.1 M cacodylate buffer. Next, they were immersed in 1% tannic acid solution, in 0.1M cacodylate buffer, for 20 minutes, and then washed 3 times for 10 min in 0.1 M cacodylate buffer. Yet again, the samples were immersed in secondary fixative for 1 hour and washed 3 times for 10 min in distilled water. Subsequently, the samples were dehydrated in a graded series of ethyl alcohol solutions (35%, 50%, and 95% v/v) for 10 minutes; then, they were washed 3 times for 10 minutes in pure ethyl alcohol. The drying was carried out to the  $\text{CO}_2$  critical point. Later, the samples were mounted on stubs using carbon tape, placing the glass slides with adhered gels onto the tape.

After mounting, the slides were sputtered with a 3-nm thick gold layer, using a sputter coater (Quorum SC7620). Lastly, the stubs were placed on the SEM sample holder (Vega 3, Tescan) for analyses.

Typical operating conditions for micrographies are with the detector of secondary electrons (SE), in high vacuum, an accelerating voltage (Acc V) of 10 for 20 kV, a working distance (WD) of 12–20 mm and a beam current of 7.0 nA.

## 2.2 Vane rheometry and the Yield Stress ( $\sigma_0$ ) of gels

The gels used for yield stress ( $\sigma_0$ ) testing were Cs1, Cs2, and Cs3, which are characterized in Table 1. The measurements were performed in a Rotational Vane Rheometer with model R/S CC (Brookfield Engineering Laboratories, Inc., Massachusetts, USA) with thermostatic bath (TC 500, Brookfield). Rotation, time, and temperature were selected, being the shearing force and rates provided by RHEO3000 software. A vane shear sensor (V-80-40 model) was used for measurements. A 200-ml aliquot was placed into the sample cup, and the temperature was raised to 40 °C for 15 minutes, with a shear rate of 0.7 RPM (shear rate of 0.23 s<sup>-1</sup>). Shortly after, the temperature was reduced to 19 °C, with no shearing force over the sample (0 RPM) for 10 minutes. Finally, the temperature of 19 °C was kept but applying a shearing force of 0.7 RPM for 90 minutes. All the essays were performed in a closed environment under exhaust hood and controlled temperature. At 40 °C, gels are at the sol stage and, when decreased to 19 °C, they pass to a gel stage. The Yield Stress ( $\sigma_0$ ) can be determined by the maximum value of the curve in the stress deformation graph [30].

## 2.3 Viscometry essays

The viscosity of 500- $\mu$ l-gel samples was measured at 37 °C in a Searle-type viscometer (Brookfield, model DV-II) coupled to a thermostatic bath circulator (TC 500, Brookfield). For this, the CPE-40 spindle was set to rotate at the shearing rates of 5.25, 7.50, 56.25, 90.00, 112.50, 150.00, 375, 750, 1,350, and 1,500 s<sup>-1</sup>. Only results with torque higher than 10% were considered. First, the samples were left to stand in the equipment for 3 minutes for thermal balance. After 30-second rotation, the measurements were performed. The arithmetic mean of the measurements was used for each shearing rate analysis.

## 2.4 Total loss testing

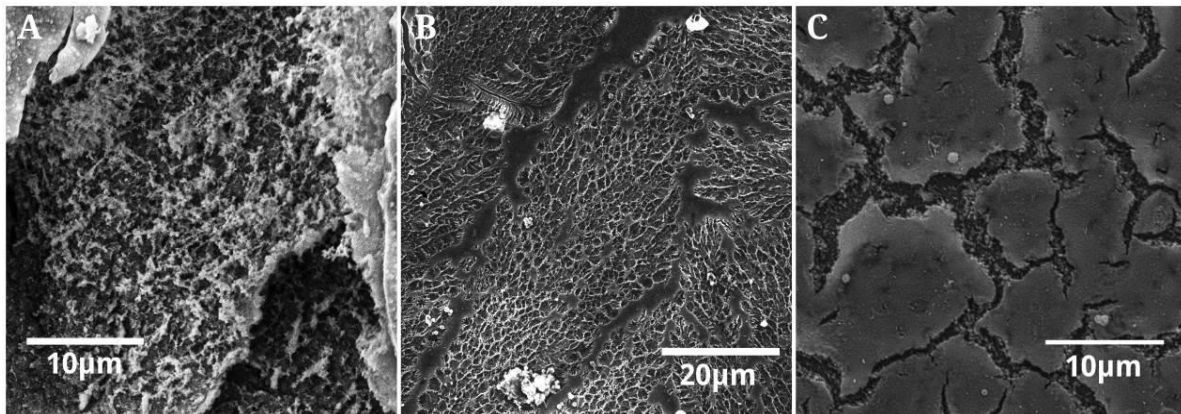
Triplicate 30-ml aliquots of collagen Cs1, Cs2, and Cs3 gel samples were weighed on an analytical scale (Shimadzu, model AUW220D, Japan) before and after trials, for a period of 12 days. The test was performed to estimate the amount of volatilized solvent during rheometric trials for gel outflow threshold (Sigma zero).

## 2.5 Differential scanning calorimetry

Gel samples were weighed ( $\pm$  15 mg) and sealed in aluminum (Al) crucibles for differential scanning calorimetry analysis using a Perkin-Elmer DSC apparatus (Perkin-Elmer, model Diamond) and a Pyris software (Perkin-Elmer). An empty Al crucible served as a reference. First, the sample was held isothermally at -10°C (minus ten degrees Celsius) for 60 seconds. Then, the samples were analyzed by a Modulated Temperature DSC technique (MTDSC) in a step scan mode (StepScan, Perkin-Elmer). Temperature programming of a step-scan is based on periodic and short temperature ranges, and on isothermal segments. The isothermal segment remains constant until the onset of a drop in heat below a predetermined value (criterion). The software provides two elements: a signal known as thermodynamic Cp - reflecting the reversible changes within the sample (fast step), and the IsoK baseline - reflecting the irreversible changes within the sample (slow step) [31, 32]. This technique was applied as follows: (a) isothermal segment – 60 seconds, (b) temperature range - 3 °C, (c) heating rate - 2 °C.min<sup>-1</sup>, and (c) balance criterion -  $\pm$  0.05  $\mu$ W. The final temperature was 40 °C, being maintained for 60 seconds. Runs of empty crucibles were used as a baseline for the measurements of heat variations related to the oven.

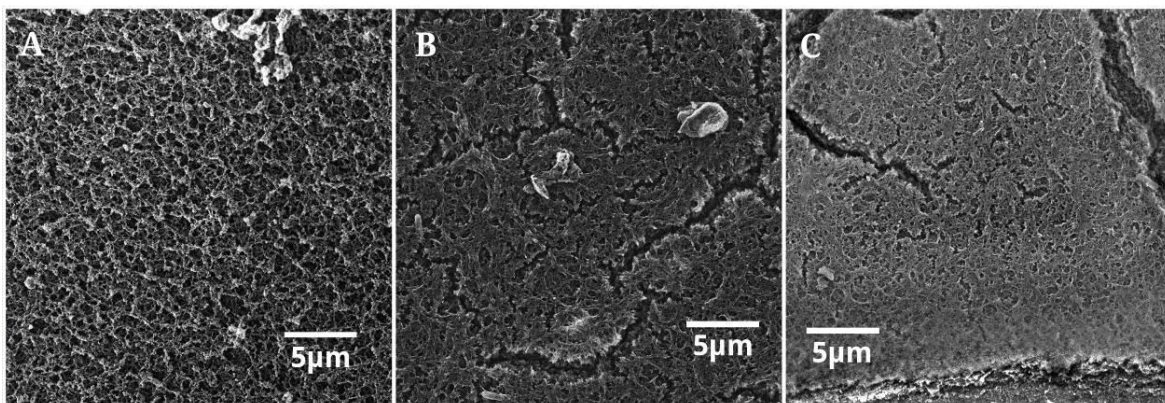
### 3. RESULTS AND DISCUSSION

SEM images of collagen gels showed differences in microstructures between gels with and without HU. In Figure 1 (A), C1 gel displays a random fiber network structure. Such constitution has fiber agglomerates at some points identified by clearer areas. Solvent retention for gel formation depends on the network formed. Conversely, in Figure 1 (B), C2 gel presents a lack of fiber networks in the collagen gel with 20mg/mL of HU; this gel also has a reduced porosity - expressed in the more compact and dense structure. Figure 1 (C) demonstrates that a rise in HU concentration from 40mg/mL HU causes a compaction in C3 gel structure; this sample undergoes strong shrinkage as can be seen by the cracks on the surface. This impairs water retention and consequently provides an increase in free water within the structure, resulting in a compact structure.



**Figure 1:** Micrographies obtained by SEM of collagen gel 6.7%, where letter A shows gel C1 without HU, letter B gel C2 with HU concentration of 20mg/ml, and letter C shows gel C3 with HU concentration of 40mg/ml. Magnification: see scale bar in the micrograph.

As shown in Figure 2 (A), the Cs1 gel without HU presents a three-dimensional fiber network matrix, which allows the trapping of solvent. In contrast, Cs4 gel sample with 20mg/ml HU (Figure 2 – B) reveals less porosity wherein its structure compresses itself (shrinkage), cracking all over the surface. Through Figure 2 (C) it was possible to observe the Cs5 gel with 40mg/ml HU has characteristics similar to Cs4 (Figure 2 - B), with cracks due to shrinkage but in a smaller number.



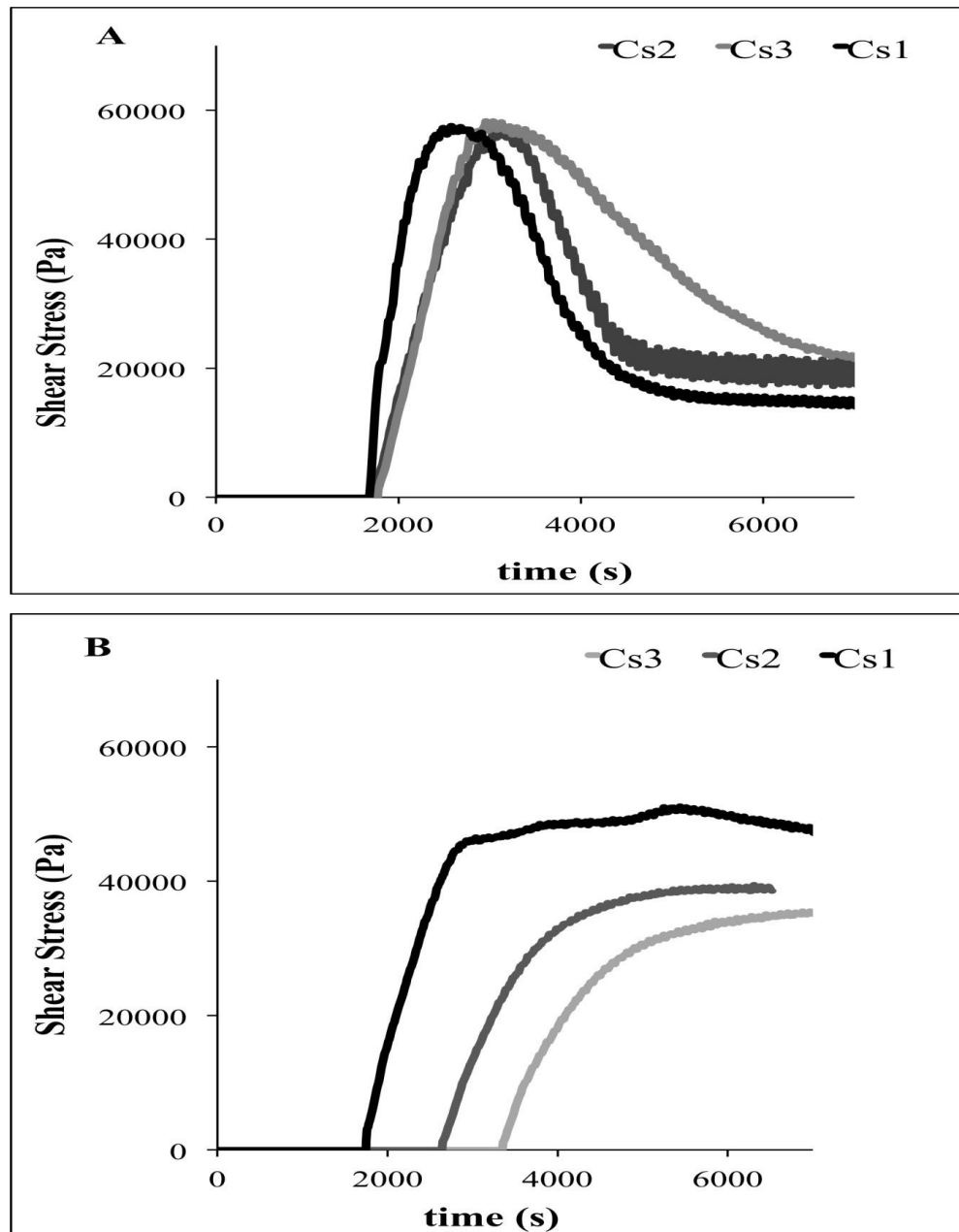
**Figure 2:** Micrograph obtained by SEM of collagen gel 2.2%, where letter A shows gel Cs1 without HU, letter B gel Cs4 with HU concentration of 20mg/ml, and letter C shows gel Cs5 with HU concentration of 40mg/ml.

Time-dependent mechanical strength tests were performed on 2.2% homogeneous gels (Cs1, Cs2, and Cs3). Figure 3(A) presents a graph of shearing force ( $\sigma$ , Pa) versus time (s) within a temperature range starting at 40 °C and ending at 19 °C (isothermal segment). The mechanical strength is similar among the gels.

The steeper the curve starts, the more rigid the gel would be [30,33]. Modulus of Rigidity (G) was determined from the slope of the curves shown in Figure 3(A) and 3(B). The value of G was higher for gel

without HU ( $G_{CS1} \sim 400$  Pa), close to the values obtained by V. Normand, et al. [34]. It also should be noted that the gel without HU reaches the yield stress ( $\sigma_0$ ) threshold with less time, which confirms the previously made statement.

The prepared gels were left at rest in a refrigerator at 10 °C. After 5 days, the same assay was repeated in the *same samples*, as shown by the graph in Figure 3(B). A change in sample rheology was then evidenced [34]. In addition to a reduction in the yield stress ( $\sigma_0$ ) threshold of all samples, those without HU showed higher values. With this, we can emphasize a decrease in stiffness of the samples as a function of time, due to the disruption of the 3D network, imposed by the application of shear stress after the gelation time (around 1 hour at temperatures below  $T_{gel}$ ). The gel does not heal after gelation has occurred so  $G$  will decrease continuously as time goes by.

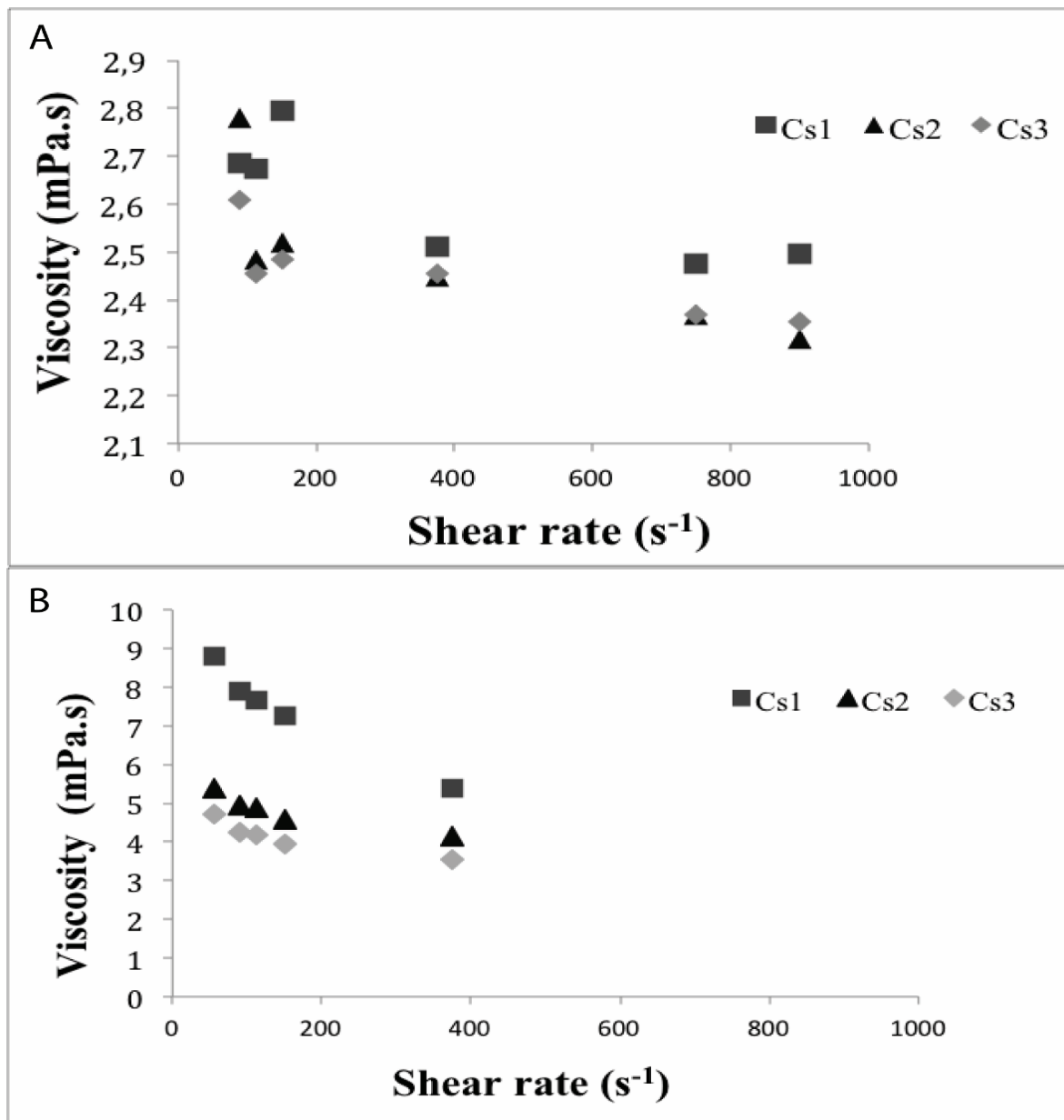


**Figure 3:** (A) Shear stress (Pa) versus gel deformation (arbitrary units) graph for Cs1 (without HU), Cs2 (0.49 mg/ml HU), and Cs3 (0.98 mg/ml HU). (B) Shear stress (Pa) versus deformation of gels Cs1 (without HU), Cs2 (0.49 mg/ml HU), and Cs3 (0.98 mg/ml HU) after 5 days.

A sol-form gel sample was used for viscosity testing, which was performed at 37 °C. Samples of 0.5 mL 2.2% gels were used in each trial. Figure 4 (A) displays the graph of viscosity ( $\eta$ ) as a function of shear-

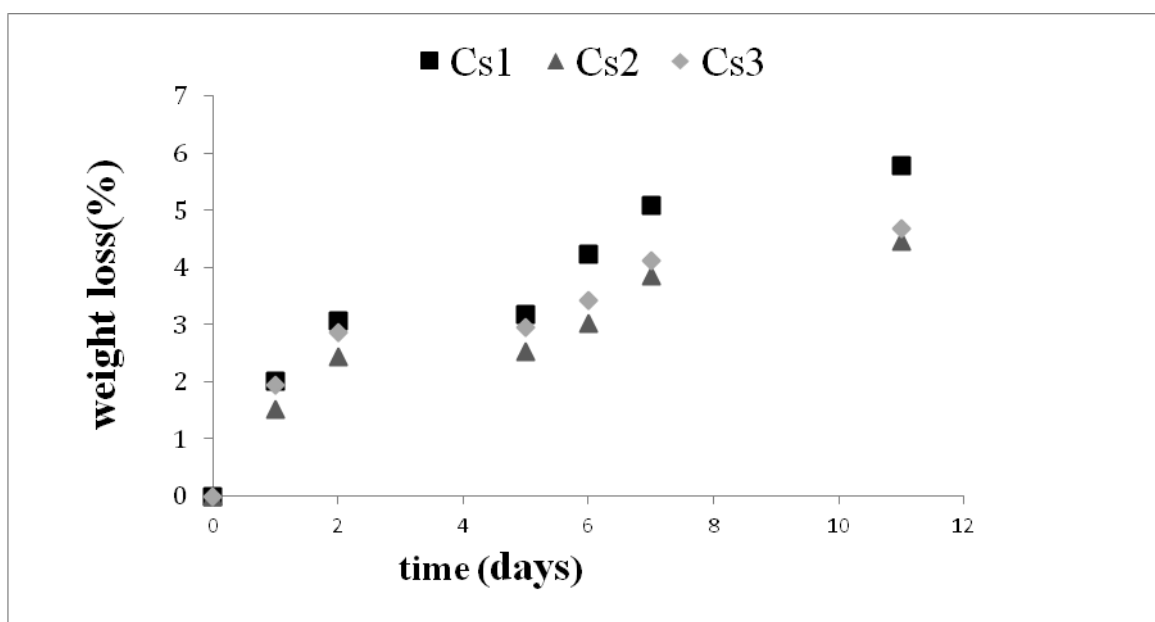
ing rates ( $\dot{\gamma}$ ) for samples without and with HU. The higher viscosity values are observed in the sample without HU, and the samples with HU have similar values among each other, as shown in the graph.

After 40 hours, another series of tests were made; the results were plotted on the graph shown in Figure 4(B). Interestingly, the viscosity increased for all samples, sample Cs1 without HU continues to have the highest values. The gel Cs3 with 40mg/ml HU had a viscosity lower than that of Cs2 with 20mg/ml HU.



**Figure 4:** (A) Viscosity versus shear rate graph of samples Cs1 (without HU), Cs2 (0.49 mg/ml HU), and Cs3 (0.98 mg/ml HU). (B) Viscosity versus shear rate graph of samples Cs1 (without HU), Cs2 (0.49 mg/ml HU), and Cs3 (0.98 mg/ml HU) after 40 hours.

A weight loss test was performed together with the viscosity testing on all gels (Cs1, Cs2, and Cs3), being made via successive heating and cooling thereof. Figure 5 demonstrates the weight loss graph according to the given treatment. For the gels with HU, Cs1 lost more mass than did the other gels. Yet the gels without HU had the highest amount of solvent within due to their three-dimensional network structure with greater retention power, as seen before in the SEM tests. The viscosity increased with time as water was lost.



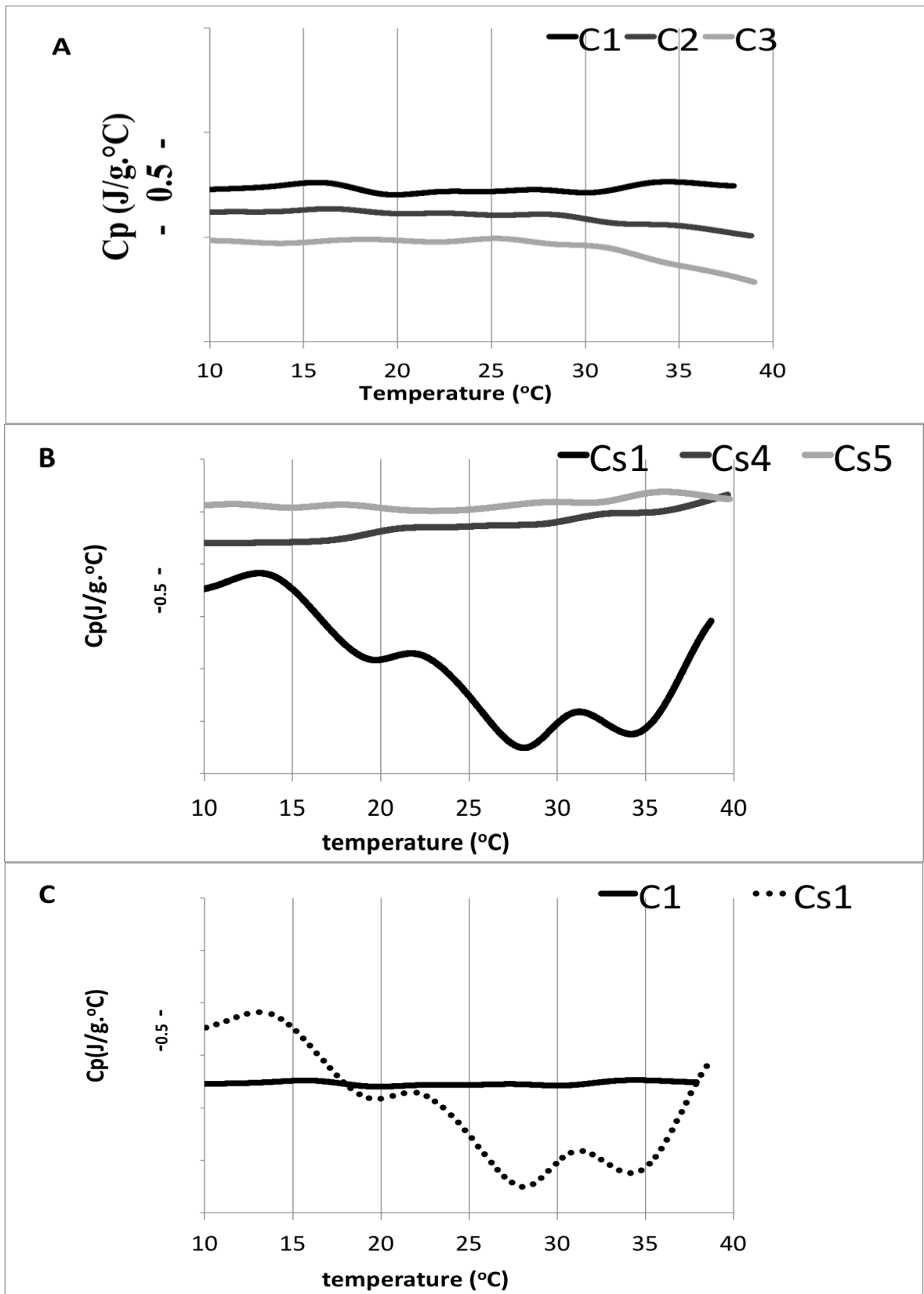
**Figure 5:** Weight loss graph by samples Cs1 (without HU), Cs2 (0.49mg/ml HU), and Cs3 (0.98mg / ml) during trials.

Figure 6(A) shows the curves obtained by the DSC plotting for the 6.7% collagen gels (C1 without HU, C2, and C3). They reveal a slight specific heat  $C_p$  difference among the gels. C1 stand out showing subtle transitions in the specific heat ( $C_p$ ) value in the temperature interval 10- 40 °C. Some of these transitions resemble glassy-state structural relaxations [35] and should be further investigated. Gels with HU do not present clearly such transitions.

On the other hand Figure 6(B) shows the thermal behavior of the 2.2% collagen gels Cs1 (without HU), Cs4 (20mg/ml HU), and Cs5 (40mg/ml), three DSC runs for each composition. Here in Figure 6(B) gels with HU have  $C_p$  values superior to that of the sample without HU. As the ratio (collagen/water) concentration is the same in each series (C and Cs), the differences in the  $C_p$  values are solely due to the physical interactions between collagen and water in the gel, whose *fiber network/water* structure is dynamic and mechanically time-dependent as shown in the other experiments of this article. This might shed some light to explain the differences between individual runs in each series.

But in both Figures 6.A and 6.B the distinction between the  $C_p$  curves of gels with and without HU are clearly seen indicating that HU alters as expected the interaction collagen/water. It also affects transitions in the collagen gel. In the case of Cs1 (no HU added) we can observe endothermic transitions around 20-25 °C and 30-35 °C. In the samples with hydroxyurea Cs2 (20mg/ml) and Cs3 (40mg/ml) these transitions are absent or not very noticeable.

From Figure 6(A), we can infer that C1 gel without HU has a better-organized three-dimensional network, which retains more solvent inside, as already seen in Figure 2(A). By contrast, gels C2 and C3, with HU, present a lower solvent retention capacity, so the  $C_p$  curve do not present structural relaxations in the temperature interval studied. Since these 6.7%-gels have a greater concentration of macromolecules and, therefore, many more hydrogen bonds increasing the intermolecular forces. By heating the sample, the average energy of molecules increases, causing them to move faster. Part of this energy is primarily used to break down these hydrogen bonds; the remainder is further transformed into molecular motion. When absorbed by the material, the solvent has its mobility restricted, thus requiring more energy to break intermolecular forces.



**Figure 6:** (A) Specific heat versus temperature by the StepScan method in collagen gels 6.7% C1 (without HU), C2 (20mg/mL), and C3 (40mg/mL). (B) Specific heat versus temperature by the StepScan method in collagen gels 2.2% Cs1 (without HU), Cs4 (20mg/mL), and Cs5 (40mg/mL). (C) Specific heat versus temperature by the DSC StepScan method indicating the differences between collagen gel 6.7% (C1) and 2.2% collagen gel (Cs1).



Collagen gels without HU are amorphous disordered elastic networks and suitable for viewing the role of slow relaxation processes on Cp at a temperature interval well below collagen denaturation. Apparently as the collagen concentration decreases the relaxations became more visible on the modulated (StepScan) DSC experiment. This assumption can also justify the comparison observed in Figure 6 (C), which shows the difference between the Cp curves of the gels C1 and Cs1 both without HU.

#### 4. CONCLUSIONS

The depolymerizing effect of hydroxyurea (HU) on bovine type I collagen gels was evidenced by several experimental techniques. Scanning electron microscopy revealed that the addition of HU results in deterioration of the three-dimensional network, typical of this gel. As is already known for quite sometime [35] the gel samples have an intense internal dynamics modifying their microstructure with time. Rheological testing pointed to a reduction in yield stress ( $\sigma_0$ ) by adding HU into collagen gels with a concentration of 2.2% w/v. In addition to a reduction in the yield stress ( $\sigma_0$ ) of all samples, those without HU showed higher values.

In the liquid phase, gels increased viscosity over time but decreased with the addition of HU, highlighting the depolymerizing effect of this substance. There was a slight weight loss in the 2.2% collagen gels Cs1 (without HU), Cs2 (0.49mg/mL), and Cs3 (0.98mg/mL), among which the gel without HU presented the highest loss. These results corroborate those obtained by SEM, where it was elucidated that most organized three-dimensional networks facilitate solvent volatilization, as in the gel without HU (Cs1).

The differential scanning calorimetry (DSC) experiments have indicated differences in the specific heat curves specially as far as the gel structural relaxations is concerned being more visible in the 2.2% w/v samples. Apparently as the collagen concentration decreases the relaxations have become more visible on the MTDSC (StepScan) experiments. Conversely, gels with HU addition showed lower solvent retention capacity, less pronounced relaxations and higher Cp values (at least for the 2.2% w/v concentration). As it is well known, as collagen in the gel increases, Cp decreases [36].

Gelatin is a physical gel, namely, its gelation is thermally reversible. The sol state ( $T > T_{gel}$ ) is a solution of single chains of denatured collagen in water. Below  $T_{gel}$  ( $\sim 27^\circ\text{C}$ ) for type I collagen, renaturation becomes thermodynamically favorable and chains form a network of segments cross-linked.

Indeed, the HU has an effect on type I collagen gelatinization probably lowering  $T_{gel}$  value by several degrees and so avoiding the 3-D network formation and gelation, being possible to consider the hypothesis that such anti-gelling effect of hydroxyurea also occurs in the polymerization of hemoglobin HbS and its reversible gelation.

From this investigative work it is suggested the extension of the study through the use of solutions containing mutant hemoglobins (sickle) to evaluate the depolymerizing effect of hydroxyurea.

#### 5. ACKNOWLEDGMENTS

We would like to thanks CAPES/MEC, Brazil, for the Master Degree scholarship to E.D.L.A and CEMIG S/A for the financial support (research project ANEEL GT343).

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